

**NANO EXPRESS**

**Open Access**

# Influence of graphene oxide on metal-insulator-semiconductor tunneling diodes

Chu-Hsuan Lin<sup>1\*</sup>, Wei-Ting Yeh<sup>1</sup>, Chun-Hui Chan<sup>1</sup> and Chun-Chieh Lin<sup>2</sup>

## Abstract

In recent years, graphene studies have increased rapidly. Graphene oxide, which is an intermediate product to form graphene, is insulating, and it should be thermally reduced to be electrically conductive. We herein describe an attempt to make use of the insulating properties of graphene oxide. The graphene oxide layers are deposited onto Si substrates, and a metal-insulator-semiconductor tunneling structure is formed and its optoelectronic properties are studied. The accumulation dark current and inversion photocurrent of the graphene oxide device are superior to the control device. The introduction of graphene oxide improves the rectifying characteristic of the diode and enhances its responsivity as a photodetector. At 2 V, the photo-to-dark current ratio of the graphene oxide device is 24, larger than the value of 15 measured in the control device.

**Keywords:** graphene oxide, MIS, tunneling diode, photodetector

## Background

Metal-oxide-semiconductor field-effect transistors (MOS-FETs) are frequently used in daily life. Strictly speaking, it is more precise to name such structures as metal-insulator-semiconductor field-effect transistors (MISFETs) since oxides represent only one class of the various insulators used today. To maximize the packing density and increase the operation speed of integrated circuits, transistor size must be reduced. The insulator thickness decreases as the dimension of a transistor becomes smaller, resulting in a significant gate tunneling current [1]. Such a tunneling current in the vertical direction (from metal to semiconductor or from semiconductor to metal) of a metal-insulator-semiconductor (MIS) structure has been employed in a number of applications, such as microwave devices [2], flash memory devices [3], solar cells [4], and photodetectors [5]. For photodetectors, the thickness of the insulator layer in the MIS tunneling diode is critical. For example, if the insulator layer is too thick, only limited tunneling can occur, leading to a small responsivity. On the contrary, if the insulator layer is too thin, Fermi level pinning may degrade the device current-voltage (IV) characteristics from rectifying behavior to ohmic behavior (or vice versa) [6,7]. In addition to the

thickness dependence, insulator composition also affects the IV characteristics of MIS devices. In this letter, we describe the effect with graphene oxide introduced in the insulator layer in MIS tunneling diodes. Graphene studies increased rapidly following the report of Novoselov et al. in 2004 describing graphene obtained from repeated peeling of graphite [8]. Many methods to prepare graphene have since been reported [9-11]. One method involves chemical oxidation of graphite followed by exfoliation of graphene oxide (GO). These GO layers are insulating and could be thermally reduced to be electrically conductive [12]. Nevertheless, the insulating property of GO may be utilized to certain applications. Each GO layer consists of the pure two-dimensional honeycomb lattice bearing functional groups, and the two-dimensional structure is not present in commonly employed insulators such as amorphous SiO<sub>2</sub>. Therefore, the influence of the GO layer in the MIS structure will be very interesting. In this study, we demonstrate MIS tunneling diodes with GO in the insulator layer. We deposited the GO layers onto the Si substrates and evaluated these structures based on the IV characteristics. We highlight the simple fabrication process employed to achieve high-performance GO-based MIS devices.

\* Correspondence: chlin0109@mail.ndhu.edu.tw

<sup>1</sup>Department of Opto-Electronic Engineering, National Dong Hwa University, Hualien 97401, Taiwan, Republic of China

Full list of author information is available at the end of the article

## Methods

The schematic process used to prepare graphene oxide on Si substrates is shown in Figure 1, and the details are given below.

### Graphite oxidation (modified Hummers method)

Graphite oxide was prepared using the modified Hummers method [13,14]. First,  $\text{H}_2\text{SO}_4$  (120 mL, 98%) was placed in an ice bath at  $< 5^\circ\text{C}$ . Then, graphite powder (5 g) and  $\text{KMnO}_4$  (15 g) was slowly added and stirred in  $\text{H}_2\text{SO}_4$  for 2 h. The temperature was controlled so as not to exceed  $35^\circ\text{C}$ . Deionized (DI) water (250 mL) was then slowly added to dilute the solution, which was then stirred for 2 h. DI water (700 mL) and  $\text{H}_2\text{O}_2$  (20 mL) were then added and stirred for 12 h. The solution was filtered and the slurry on the filter paper was washed with HCl (3%) to remove inorganic impurities. The slurry was then washed with DI water (1 L) to remove the residual acid. Finally, the slurry was dried to yield graphene oxide.

### GO suspension

Graphite oxide powder (30 mg) was added to DI water (20 mL). After stirring for 10 min, ultrasonication was performed for 2 h to exfoliate the GO sheets from the multi-layer flakes. The solution was then centrifuged at 5,000 rpm for 15 min, and the supernatant was collected. Due to the polar oxygen-containing functional groups, GO sheets were well suspended in the supernatant [15], which was subsequently used for dip-coating.

### Treatment on Si and GO dip-coating

In this study, 1- to  $10\text{-}\Omega\text{ cm}$  p-type Si with native oxide was used as the substrates. The substrate was first cleaned and made hydrophilic by treatment with SC1

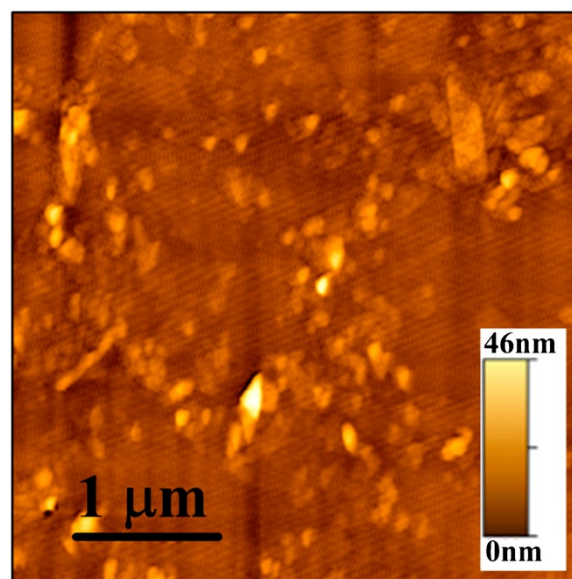
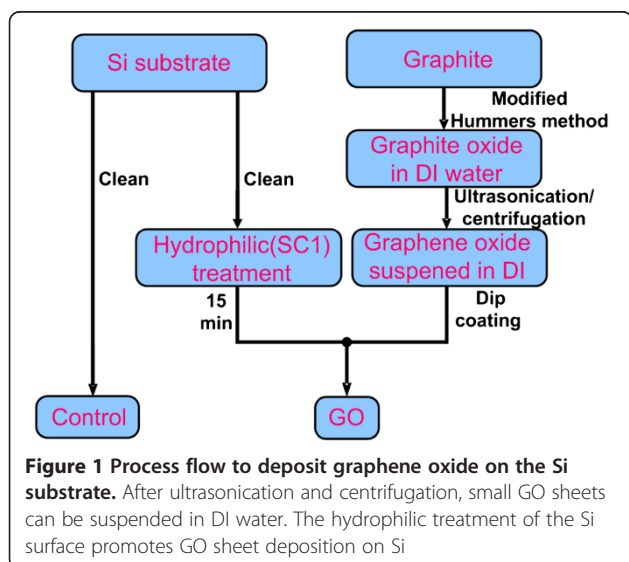
solution ( $\text{NH}_4\text{OH}/\text{H}_2\text{O}_2/\text{H}_2\text{O} = 1:2:8$ ) for 15 min. The SC1 treatment rendered the surface hydrophilic by attaching polar hydroxyl groups as described in [16]. The GO sheets were then deposited on substrates by dip-coating. The GO deposition was driven by the van der Waals interaction between the hydrophilic substrate and the polar oxygen-containing groups of the GO. Following dip-coating for 40 min, the dried substrate was covered with multilayers of GO sheets. Each single GO sheet might be approximately 1 nm thick as described in [12]. However, the method of chemical exfoliation resulted in multiple-sheet deposition. From the profile of the atomic force microscopy (AFM) image, the thickness of the GO film in the GO device was estimated to be approximately 10 nm.

### Device fabrication

A 100-nm-thick Al gate with a circular area of  $5 \times 10^{-4}\text{ cm}^2$  defined by a shadow mask was sputtered onto the GO-deposited side of the Si substrate. Al was also large-scale sputtered onto the back side of the Si substrate to form an ohmic contact. In order to identify the influence of the GO layer on the device characteristics, we also prepared and evaluated a control device without a GO film. White-light LEDs operating at a power density of  $0.5\text{ mW/cm}^2$  were the illumination source used for photocurrent measurements.

## Results and discussion

We performed AFM to investigate the morphology of the GO sheets on the Si substrate (Figure 2). The root mean square roughness was 3.1 nm. It seemed that only the



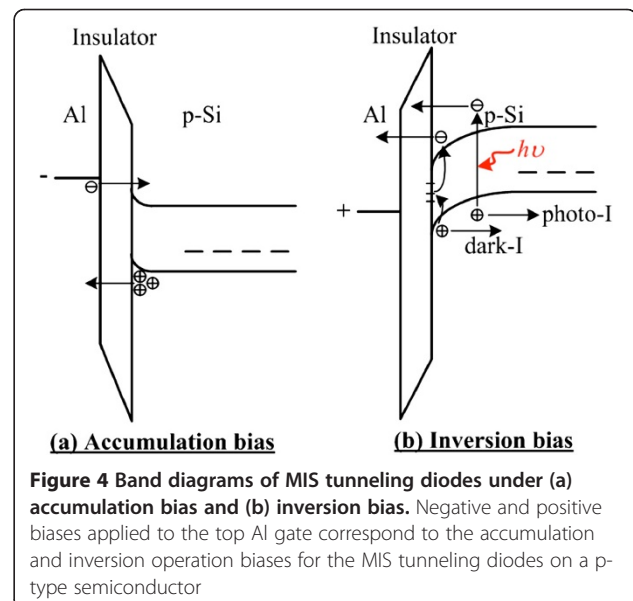
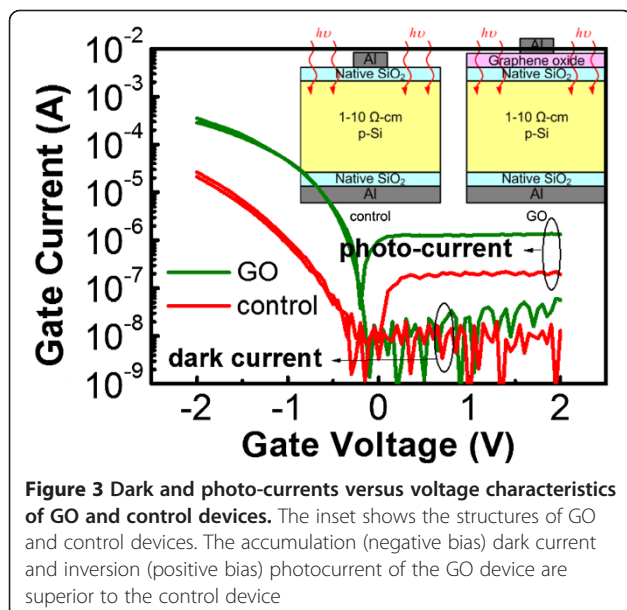
**Figure 2** AFM image of GO on the Si substrate. The sample seems to be only partially covered with GO flakes

partial area was covered with GO flakes. The dark and photo-currents versus voltage characteristics of GO and control devices are shown in Figure 3, and the structures of GO and control devices are shown in the inset of Figure 3. The basic operation principle of an MIS photodetector can be found in [5]. Details regarding the conduction mechanisms in MIS tunneling diodes under an accumulation bias have been discussed in [17], and mechanisms under an inversion bias have been discussed in [18,19]. The tunnel diode is broadly used in low-power microwave applications. However, the MIS tunnel structure is also promising for use as a photodetector. As reviewed in [5], MIS tunneling diodes can be used to detect ultraviolet, visible, and infrared light. Photodetectors based on the MIS structure could own a smaller dark current than metal-semiconductor-metal photodetectors. The fabrication process is also simpler than that of pn photodetectors. Figure 4a,b shows the band diagrams under the accumulation bias and the inversion bias, respectively. Negative and positive biases applied to the top Al gate correspond to the accumulation and inversion operation biases, respectively, for the MIS tunneling diodes on a p-type semiconductor. At accumulation (negative) biases, holes are accumulated at the insulator/semiconductor interface and tunnel to the top gate. In addition, the Fermi level of the top gate continuously rises above the conduction band edge of the semiconductor. Hence, electrons can also tunnel from the top gate to the semiconductor (Figure 4a). The magnitude of the accumulation current is therefore related to the magnitude of the applied bias voltage (Figure 3). The magnitude of the inversion current is less dependent on the magnitude of the bias voltage. It is because that the inversion dark current

is mainly contributed by thermally generated electron-hole pairs at the insulator/semiconductor interface and bulk traps in the depletion region. When the devices are irradiated, extra photo-generated electron-hole pairs result in a photocurrent (Figure 4b).

From the IV measurements, it is evident that the dark current of the GO device at the accumulation bias (for example, at  $-2$  V) is larger than that of the control device. Since the magnitudes of the inversion (for example, at  $2$  V) dark currents of both devices are similar, the GO device has improved rectifying characteristics. At the inversion bias, the photocurrent of the GO device is also larger than that of the control device, i.e., at  $2$  V, the photocurrent of the GO MIS tunneling diode is  $1.34 \times 10^{-6}$  A, while that of the control diode is  $1.94 \times 10^{-7}$  A. This shows that the use of GO can increase the responsivity of MIS photodetectors. To show the characteristic variation within different devices, the cumulative probability plot of the collected data is shown in Figure 5. There are nine samples used both for the GO and control samples. In the cumulative probability plot, the formula of  $i/n$  is used for the cumulative probability calculation of the y-axis (in percentage), where  $i$  is the rank number and  $n$  is the number of data points. The presence of GO layers can indeed increase the accumulation ( $-2$  V) dark currents and inversion ( $2$  V) photocurrents measured in MIS devices.

At  $2$  V, the photo-to-dark current ratio of the GO device is 24 compared to the value of 15 obtained for the control device. The photocurrent of the GO device is much larger than that of the control device due to the introduction of GO. Meanwhile, the dark current of the GO device is only slightly increased. Overall, the GO device can achieve a better photo-to-dark current ratio.



In [18], the photocurrent density of an MIS tunneling diode is approximately  $0.03 \text{ mA/cm}^2$  under  $0.71\text{-mW/cm}^2$  illumination. The photocurrent density of our GO MIS tunneling diode is approximately  $3 \text{ mA/cm}^2$  under  $0.5\text{-mW/cm}^2$  illumination; therefore, a much higher responsivity could be achieved via GO incorporation. Although the smaller dark current of the diode reported in [18] would result in a better photo-to-dark current ratio, the photo-to-dark current ratio of our GO device could increase if the GO is deposited on a high-quality thermal oxide as opposed to the native oxide.

The beneficial influence of GO may be attributed to its negative interface charge (or oxide charge).  $\text{Al}_2\text{O}_3$  has been previously used to passivate the surface of p-Si due to the negative charge at its interface, and the capacitance-voltage curves shift toward positive voltage when more negative charge is present [20]. We have measured the capacitance-voltage relation of the GO and control samples. Although the negative-bias data could not be obtained due to the large accumulation currents of our devices, the positive-bias data show a positive shift similar to the  $\text{Al}_2\text{O}_3$  case (Figure 6). If there is negative charge at the interface, holes prefer to accumulate at the surface of p-Si. Minority carriers, electrons, could be pushed away from the interface of p-Si, and it leads to a low surface recombination [20]. Low surface recombination could contribute to the high inversion photocurrent of the GO sample. On the other hand, the inherent hole accumulation present at the surface of p-Si of the GO sample also results in the larger accumulation currents observed at the negative bias.

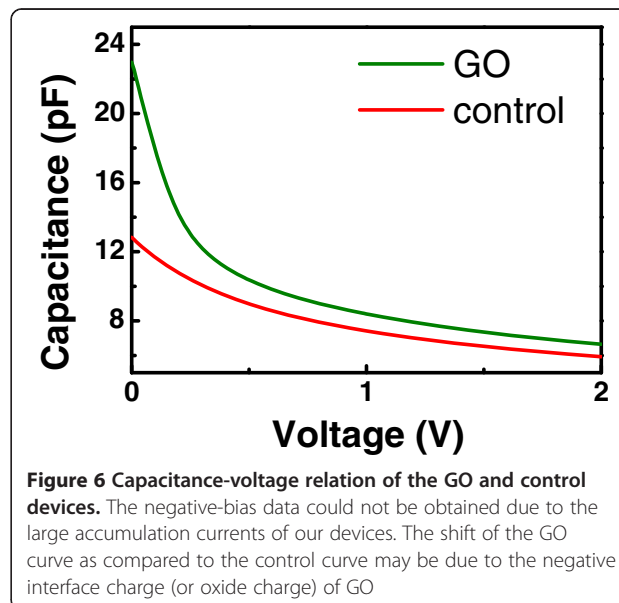
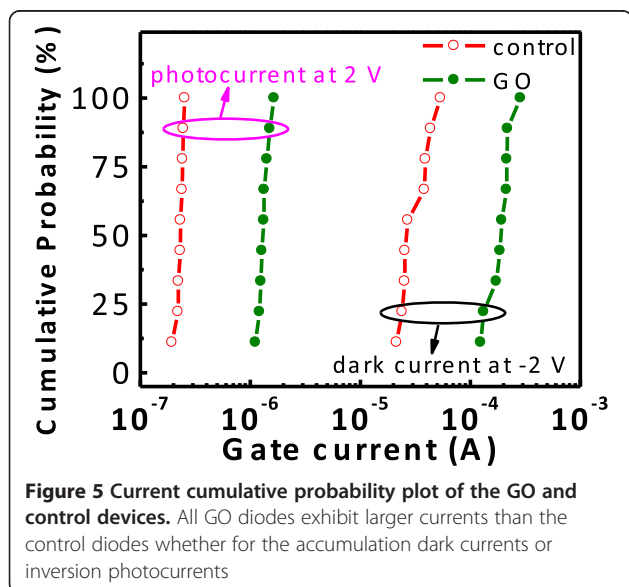
Although thick GO flakes may prevent the tunneling of carriers, carriers can still flow through the native oxide that it is not covered with thick GO flakes for the GO sample. In other words, the GO sample is only partially covered

with GO, and carriers can tunnel via the uncovered regions.

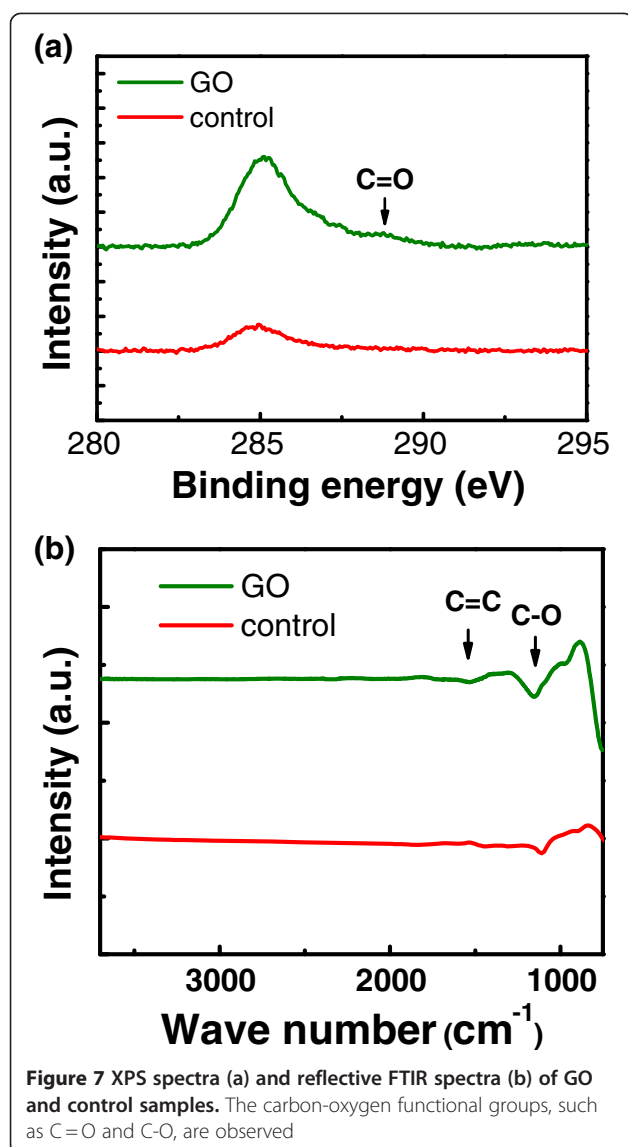
We used X-ray photoelectron spectroscopy (XPS) and reflective Fourier transform infrared spectroscopy (FTIR) to identify the chemical composition of the GO films. Figure 7a shows the carbon (C) 1s XPS spectra of GO and control samples. The peak at approximately 285 eV corresponds to C 1s peak of an  $sp^2$  C or  $sp^3$  C [21], and the small peak at approximately 289 eV corresponds to a carbonyl carbon ( $\text{C}=\text{O}$ ) [22]. Since the control sample has been cleaned in acetone and methanol, a small C 1s peak at approximately 285 eV can be observed. Figure 7b shows the reflective FTIR spectra of GO and control samples. The GO spectrum is obtained with the Si substrate used in background measurement; hence, the final spectrum only reveals the reflection characteristics of the graphene-oxide film. In addition to the absorption due to C=C bonds, the absorption due to C-O bonds can also be observed. Similar results have been observed in [23]. The 'carbon-oxygen' functional groups are responsible for the insulating properties of GO. We attempted to perform Hall measurement to obtain the resistivity and mobility of the GO films; however, since the top layers were insulators instead of semiconductors, no reliable electrical data could be obtained.

## Conclusions

Graphene oxide was deposited on Si to form an Al/graphene oxide/native oxide/Si MIS tunneling diode. With GO insertion, the accumulation dark current and inversion photocurrent are greater than those measured in the control device. The photocurrent of the GO MIS tunneling diode is  $1.34 \times 10^{-6} \text{ A}$ , while the photocurrent of the control diode is  $1.94 \times 10^{-7} \text{ A}$ ; thus, the GO-based device is a







**Figure 7** XPS spectra (a) and reflective FTIR spectra (b) of GO and control samples. The carbon-oxygen functional groups, such as C=O and C-O, are observed

promising candidate for detector applications. Future work can involve optimization of GO coverage in MIS tunneling diodes.

#### Competing interests

The authors declare that they have no competing interests.

#### Acknowledgements

The authors are grateful to the National Nano Device Laboratories of the Republic of China for the facility access and the National Center for High-Performance Computing of the Republic of China for the computer time and facility access. This work was supported by the National Science Council of the Republic of China under contract no. 98-2221-E-259-002-MY3.

#### Author details

<sup>1</sup>Department of Opto-Electronic Engineering, National Dong Hwa University, Hualien 97401, Taiwan, Republic of China. <sup>2</sup>Department of Electrical Engineering, National Dong Hwa University, Hualien 97401, Taiwan, Republic of China.

#### Authors' contributions

CHL conceived the experimental methodology, modified the process flow, and prepared the manuscript. WTY and CHC performed the device fabrication and measurement. CCL assisted with the device fabrication and participated in the modification of the methodology. All authors read and approved the final manuscript.

#### Authors' information

CHL is an assistant professor in the Department of Opto-Electronic Engineering, National Dong Hwa University. WTY and CHC are currently working toward their master degree in the Department of Opto-Electronic Engineering, National Dong Hwa University. CCL is an assistant professor in the Department of Electrical Engineering in the same school.

Received: 30 November 2011 Accepted: 26 June 2012

Published: 1 August 2012

#### References

1. Lee WC, Hu C: Modeling CMOS tunneling currents through ultrathin gate oxide due to conduction- and valence-band electron and hole tunneling. *IEEE Trans Electron Devices* 2001, **48**:1366–1373.
2. Sze SM, Ng KK: *Physics of Semiconductor Devices*. 3rd edition. Hoboken: Wiley; 2007.
3. Seo YJ, Kim KC, Kim TG, Sung YM, Cho HY, Joo MS, Pyi SH: Analysis of electronic memory traps in the oxide-nitride-oxide structure of a polysilicon-oxide-nitride-oxide-semiconductor flash memory. *Appl Phys Lett* 2008, **92**:132104.
4. Har-Lavan R, Ron I, Thieblemont F, Cahen D: Toward metal-organic insulator-semiconductor solar cells, based on monolayer self-assembly on n-Si. *Appl Phys Lett* 2009, **94**:043308.
5. Lin CH, Liu CW: Metal-insulator-semiconductor photodetectors. *Sensors* 2010, **10**:8797–8826.
6. Kobayashi M, Kinoshita A, Saraswat K, Wong HSP, Nishi Y: Fermi level depinning in metal/Ge Schottky junction for metal source/drain Ge metal-oxide-semiconductor field-effect-transistor application. *J Appl Phys* 2009, **105**:023702.
7. Lieten RR, Degroote S, Kuijk M, Borghs G: Ohmic contact formation on n-type Ge. *Appl Phys Lett* 2008, **92**:022106.
8. Novoselov KS, Geim AK, Morozov SV, Jiang D, Zhang Y, Dubonos SV, Grigorieva IV, Firsov AA: Electric field effect in atomically thin carbon films. *Science* 2004, **306**:666–669.
9. Camara N, Jouault B, Jabakhanji B, Caboni A, Tiberj A, Consejo C, Godignon P, Camassel J: Multidimensional characterization. Landau levels and density of states in epitaxial graphene grown on SiC substrates. *Nanoscale Res Lett* 2011, **6**:141.
10. Enderlein C, Kim YS, Bostwick A, Rotenberg E, Horn K: The formation of an energy gap in graphene on ruthenium by controlling the interface. *New J Phys* 2010, **12**:033014.
11. Lee BJ, Yu HY, Jeong GH: Controlled synthesis of monolayer graphene toward transparent flexible conductive film application. *Nanoscale Res Lett* 2010, **5**:1768–1773.
12. Eda G, Fanchini G, Chhowalla M: Large-area ultrathin films of reduced graphene oxide as a transparent and flexible electronic material. *Nat Nanotechnol* 2008, **3**:270–274.
13. Hummers WS, Offeman RE: Preparation of graphitic oxide. *J Am Chem Soc* 1958, **80**:1339.
14. Becerril HA, Mao J, Liu Z, Stoltenberg RM, Bao Z, Chen Y: Evaluation of solution-processed reduced graphene oxide films as transparent conductors. *ACS Nano* 2008, **2**:463–470.
15. Gilje S, Han S, Wang M, Wang KL, Kaner RB: A chemical route to graphene for device applications. *Nano Lett* 2007, **7**:3394–3398.
16. Lin CH, Yang YJ, Encinas E, Chen WY, Tsai JJ, Liu CW: Single crystalline film on glass for thin film solar cells. *J Nanosci Nanotechnol* 2009, **9**:3622–3626.
17. Lin CF, Liu CW, Chen MJ, Lee MH, Lin IC: Electroluminescence at Si band gap energy based on metal-oxide-silicon structures. *J Appl Phys* 2000, **87**:8793–8795.
18. Liu CW, Liu WT, Lee MH, Kuo WS, Hsu BC: A novel photodetector using MOS tunneling structures. *IEEE Electron Devices Lett* 2000, **21**:307–309.
19. Hsu BC, Chang ST, Chen TC, Kuo PS, Chen PS, Pei Z, Liu CW: A high efficient 820 nm MOS Ge quantum dot photodetector. *IEEE Electron Devices Lett* 2003, **24**:318–320.

20. Agostinelli G, Delabie A, Vitanov P, Alexieva Z, Dekkers HFW, Wolf SD, Beaucarne G: **Very low surface recombination velocities on p-type silicon wafers passivated with a dielectric with fixed negative charge.** *Solar Energy Materials & Solar Cells* 2006, **90**:3438–3443.
21. Mérel P, Tabbal M, Chaker M, Moisa S, Margot J: **Direct evaluation of the  $sp^3$  content in diamond-like-carbon films by XPS.** *Appl Surf Sci* 1998, **136**:105–110.
22. Dimiev A, Kosynkin DV, Sinitskii A, Slesarev A, Sun Z, Tour JM: **Layer-by-layer removal of graphene for device patterning.** *Science* 2011, **331**:1168–1172.
23. Li X, Zhang G, Bai X, Sun X, Wang X, Wang E, Dai H: **Highly conducting graphene sheets and Langmuir-Blodgett films.** *Nature Nanotechnology* 2008, **3**:538–542.

doi:10.1186/1556-276X-7-343

**Cite this article as:** Lin et al.: Influence of graphene oxide on metal-insulator-semiconductor tunneling diodes. *Nanoscale Research Letters* 2012 **7**:343.

**Submit your manuscript to a SpringerOpen<sup>®</sup> journal and benefit from:**

- Convenient online submission
- Rigorous peer review
- Immediate publication on acceptance
- Open access: articles freely available online
- High visibility within the field
- Retaining the copyright to your article

---

Submit your next manuscript at ► [springeropen.com](http://springeropen.com)